

framatome

Modeling of thermal
expansion and
metallurgical phases of a
material during its cooling

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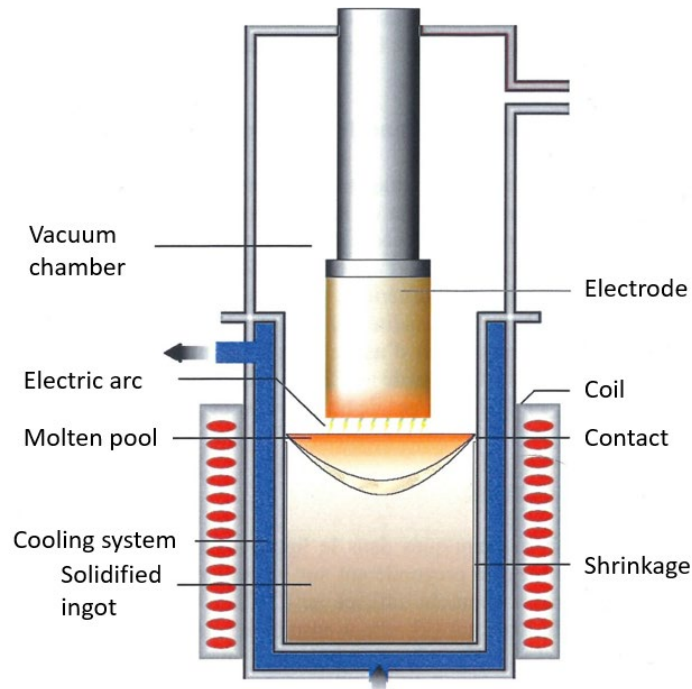
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Context

Context

Foundry Processes

- Various melting processes



A schematic diagram of the VAR process

*"Modélisation mathématique et simulation numérique du procédé de refusion à arc sous vide",
Alain Jardy, 2005*

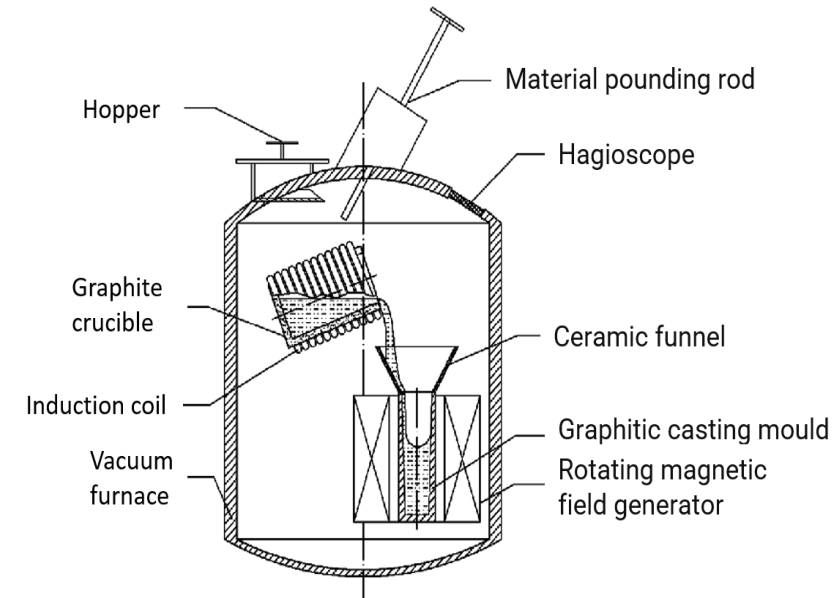
Vacuum Arc
Remelting
(VAR)

Vacuum Induction
Melting
(VIM)

To predict the resulting metallurgy



Understand and model precisely
thermal exchanges during the process



A schematic diagram of the VIM process.

« The effect of rotating magnetic field on the microstructure of in situ TiB₂/Cu composites. » Zou, Cunlei & Kang, Huijun & Li, Rengeng & Li, M & Wang, Wei & Chen, Zongning & Wang, Tongmin. (2016). IOP Conference Series: Materials Science and Engineering. 117. 012043. 10.1088/1757-899X/117/1/012043.

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Modeling

Modeling

Thermal Exchanges & Deformed Geometry

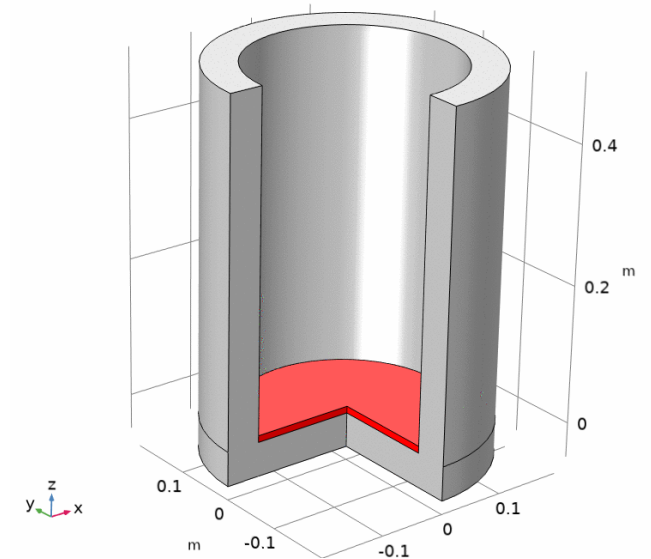
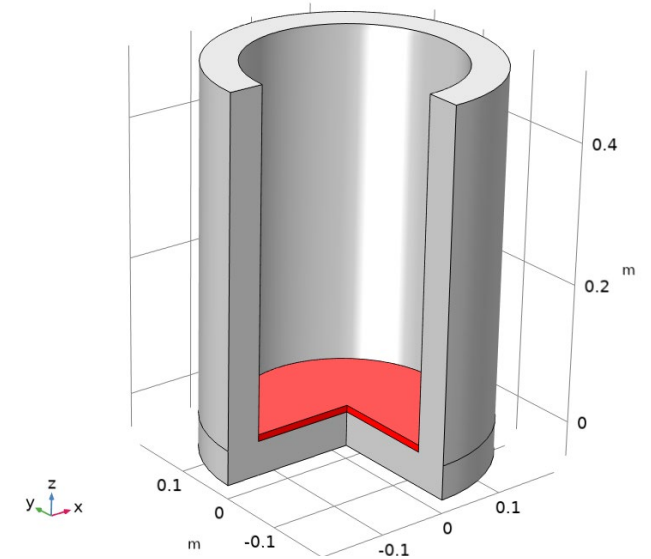
- 2D axisymmetric assumption
- **Heat equation** solved in a **deformed geometry** to model the growth of the ingot during mold filling

$$\rho C_p \frac{\partial T}{\partial t} + \nabla \cdot (-k \nabla T) = 0$$

- Filling Velocity is known analytically, and **Mesh Deformation is prescribed**

$$\Delta Z_{top} = \frac{(v_{ingot} \cdot t \cdot f(t) + h_{ingot}(1 - f(t))) \cdot Z}{h_{ingot}}$$

- Temperature of the top boundary of the ingot is maintained at $T = T_{melt} + 100$ during the filling with a heat flux condition



Modeling

Thermal Exchanges – Boundary Conditions

- Thermal Contact
$$-\mathbf{n}_1 \cdot \mathbf{q}_1 = -h_{ingot}(T_2 - T_1)$$
$$-\mathbf{n}_2 \cdot \mathbf{q}_2 = -h_{ingot}(T_1 - T_2)$$

- Three zones :

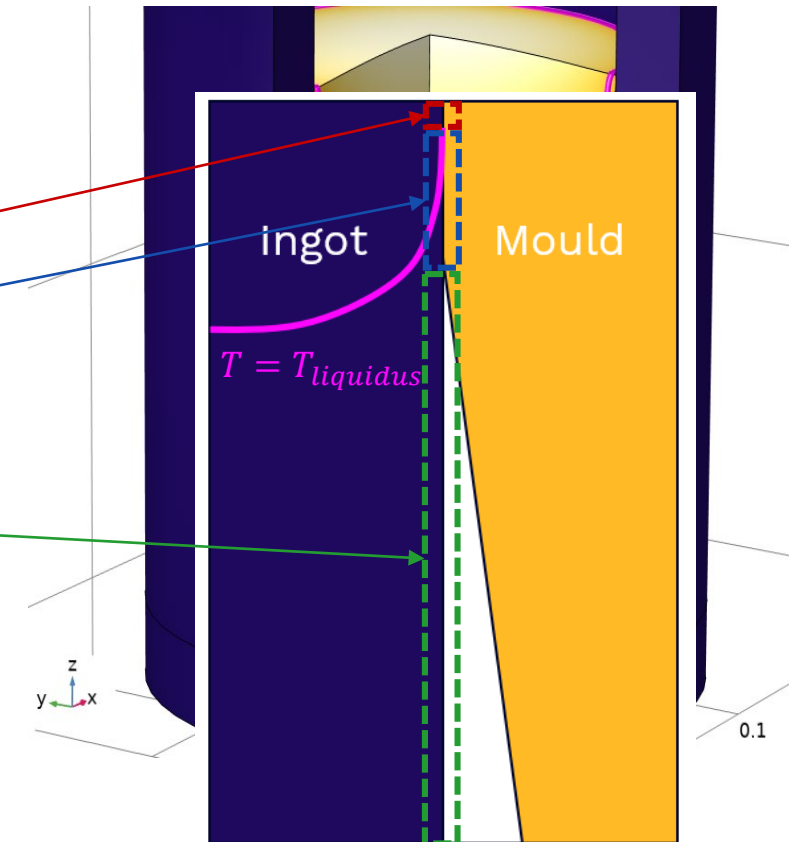
$$h_{ingot} = 10^4 \frac{W}{m^2 \cdot K} \text{ if } T_1 > T_{liquidus}$$

$$h_{ingot} = 800 \frac{W}{m^2 \cdot K} \text{ if contact}$$

$$h_{ingot} = \varepsilon_1 \sigma (T_1^2 + T_2^2) (T_1 + T_2) \text{ if no contact}$$

- Cooling conditions
$$-\mathbf{n} \cdot \mathbf{q} = h_{cooling}(T_{fluid} - T)$$

$h_{cooling}$ obtained with Comsol Multiphysics relations



Modeling

Metallurgy

- **Austenite to ferrite and pearlite** (diffusion behavior):

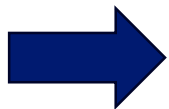
$$\frac{dz_i}{dt} = K_i(T)z_{austenite} - L_i(T)z_i$$

- Austenite to **bainite** (diffusion behavior but function of the thermal kinetics)

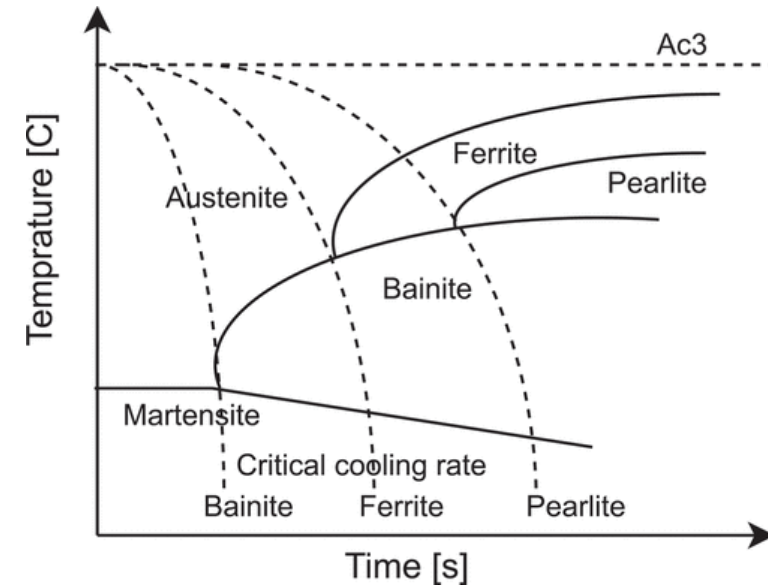
$$\frac{dz_{bainite}}{dt} == F(T)H\left(\frac{dT}{dt}\right)z_{austenite} - G(T)H\left(\frac{dT}{dt}\right)z_{bainite}$$

- **Martensite** formation if $\frac{dT}{dt} < 0$ and $T < M_s$

$$\frac{dz_{martensite}}{dt} == -z_{austenite}\beta \frac{dT}{dt}$$



4 Distributed ODEs are added



A CCT diagram for steel

"Prediction of continuous cooling transformation diagram for weld heat affected zone by machine learning", Satoshi Minamoto, 2022

Modeling

Mechanics

- Equation $\nabla \cdot \underline{\underline{\sigma}} = \mathbf{0}$

- Lemaitre & Chaboche [*] model

$$\underline{\underline{\sigma}} = \underline{\underline{\mathbb{E}}}(T) : \left(\underline{\underline{\varepsilon}} - \underline{\underline{\varepsilon}}^p - \alpha(T, z_i) \cdot (T - T_0) \underline{\underline{\mathbf{I}}} \right)$$

$$\underline{\underline{\varepsilon}} = \frac{1}{2} (\nabla \mathbf{u} + \nabla \mathbf{u}^T)$$

$$\underline{\underline{\varepsilon}}^p = \dot{p} \frac{3}{2} \frac{\underline{\underline{\sigma}}' - \underline{\underline{X}}'}{J_2(\underline{\underline{\sigma}}' - \underline{\underline{X}}')}

$$\underline{\underline{\dot{X}}} = \frac{2}{3} C(T) \underline{\underline{\varepsilon}}^p - \gamma \underline{\underline{X}} \dot{p}$$

$$\dot{p} = \left\langle \frac{J_2(\underline{\underline{\sigma}} - \underline{\underline{X}}) - \sigma_y(T)}{K(T)} \right\rangle^{n(T)}$$$$

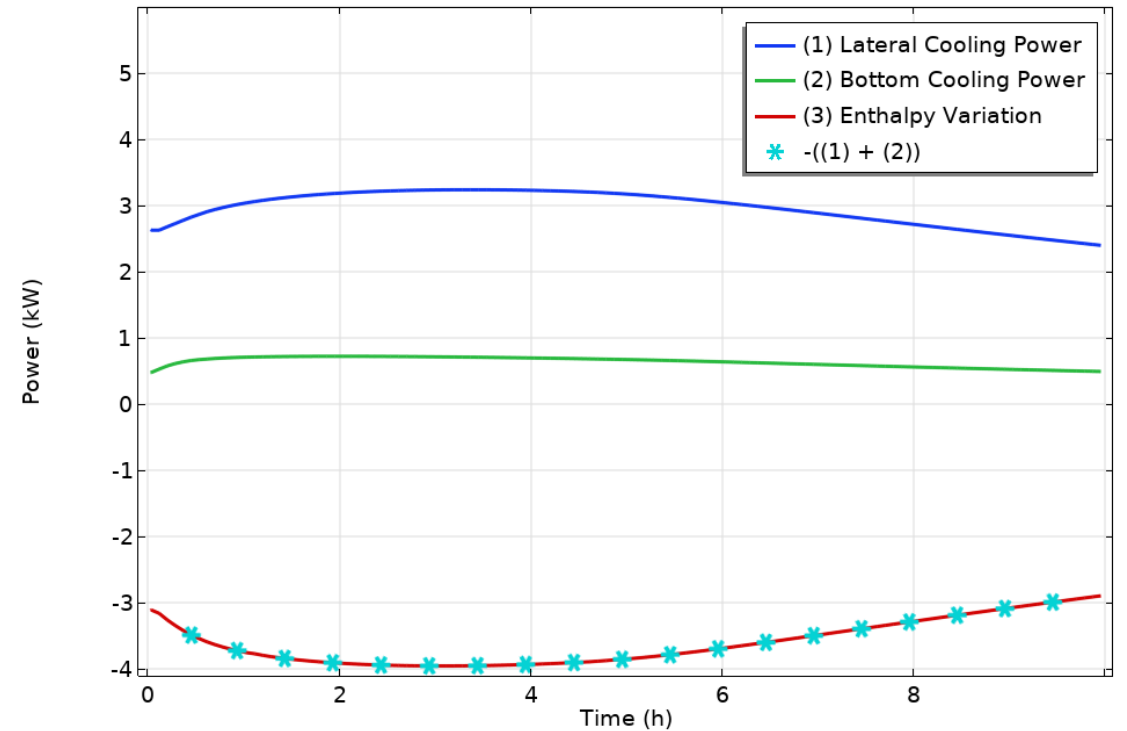
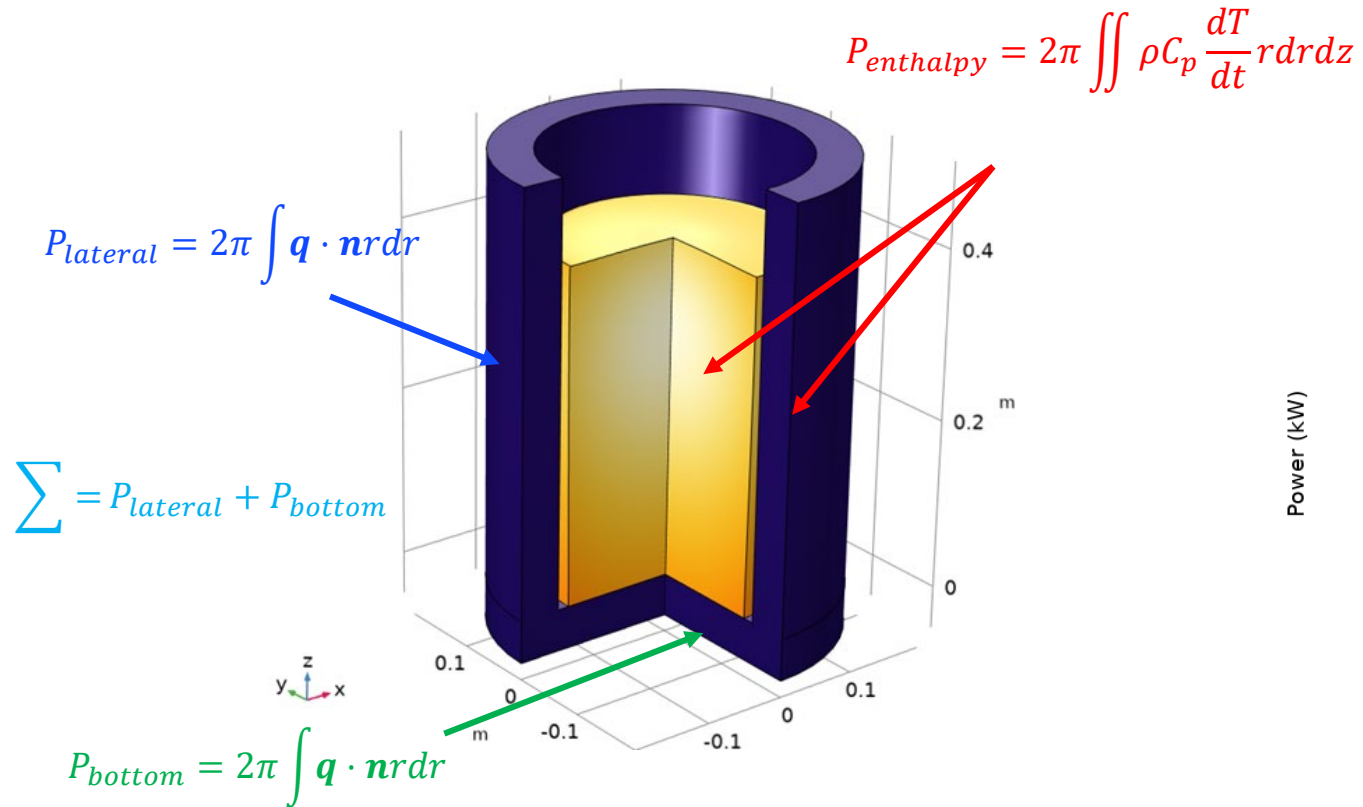
[*] J. Lemaitre et J.-L. Chaboche, Mécanique des matériaux solides, Dunod, 2001.

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Numerical Aspects & Validation

Numerical Validation

Energy Balance



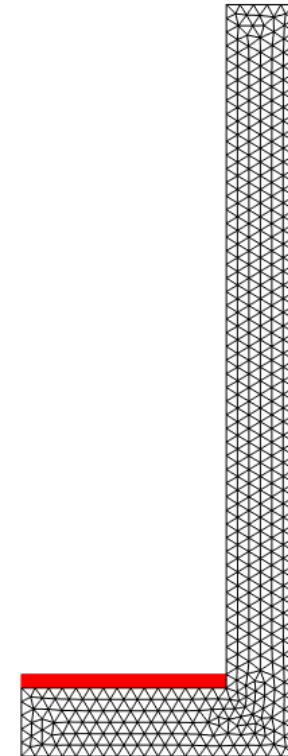
- Numerical Validation with energy balance verifications

Numerical Aspects

Mesh and Solvers

- Mapped Mesh
- Refinement near the lateral contact conditions
- Deformation analytically controlled

- Segregated Resolution Technique
- Time-Step control

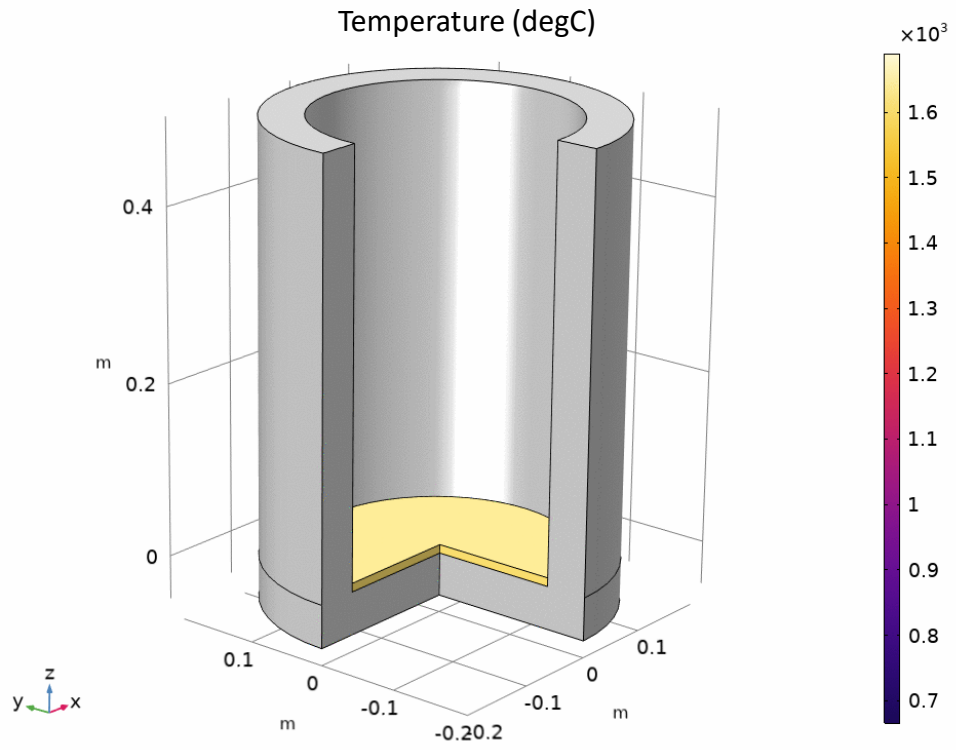


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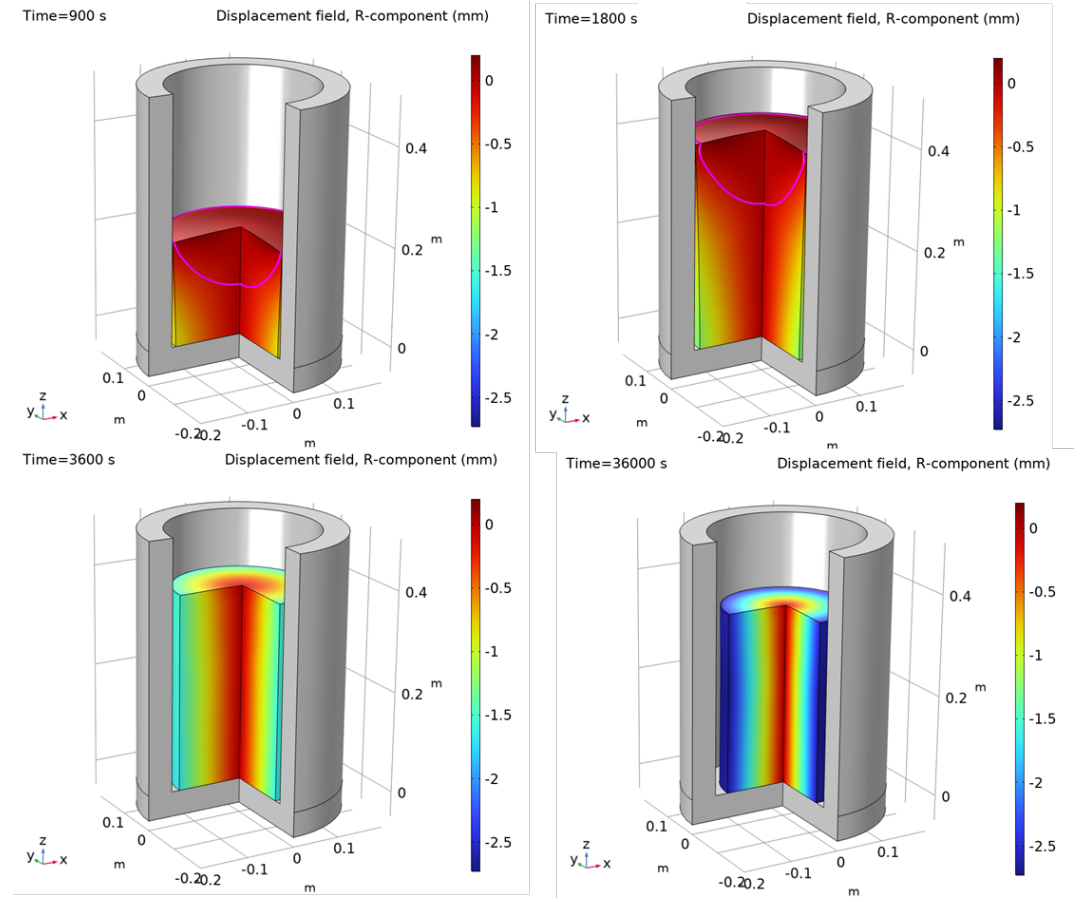
Results

Results

Thermo-Mechanical Results

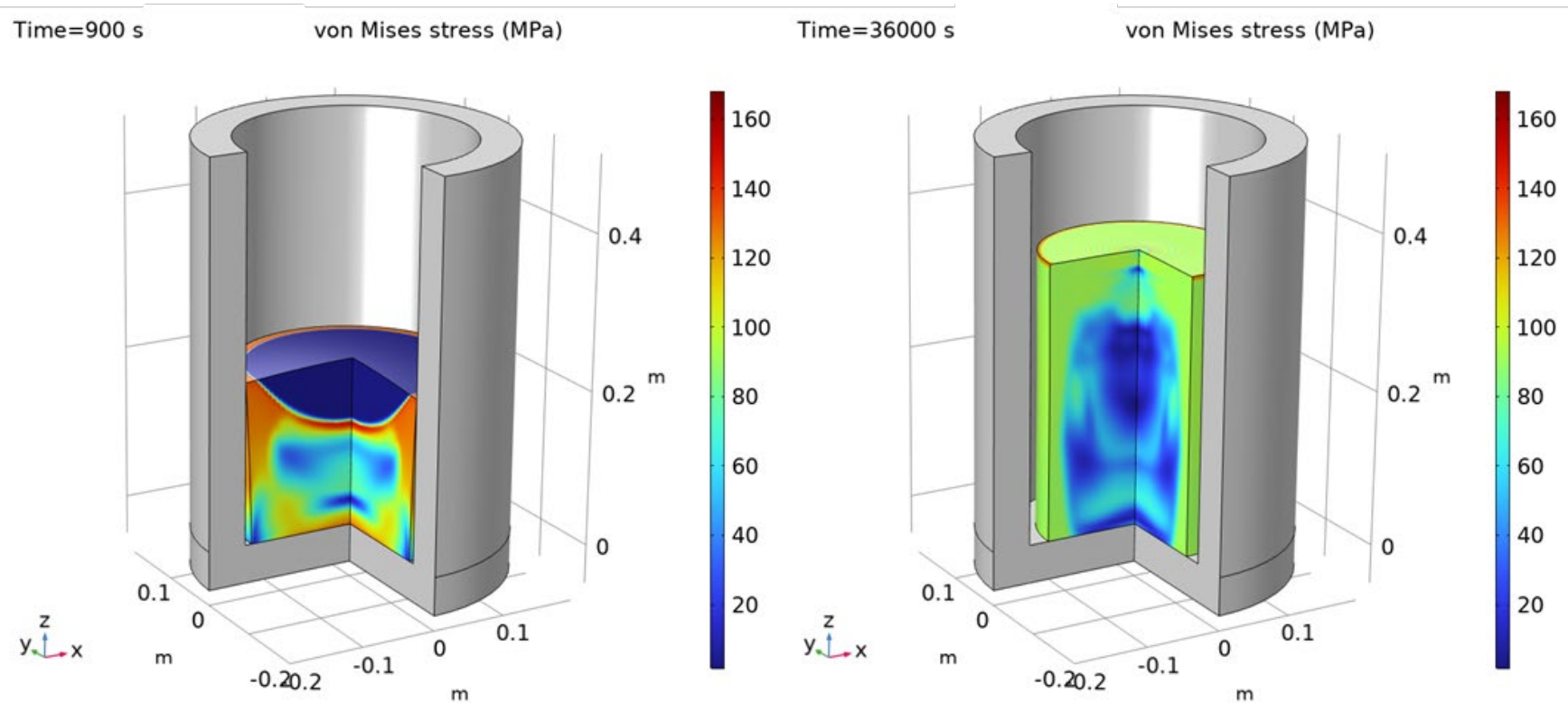


- Filling and Cooling phases can be studied
- Temperature field and thermal shrinkage are predicted



Results

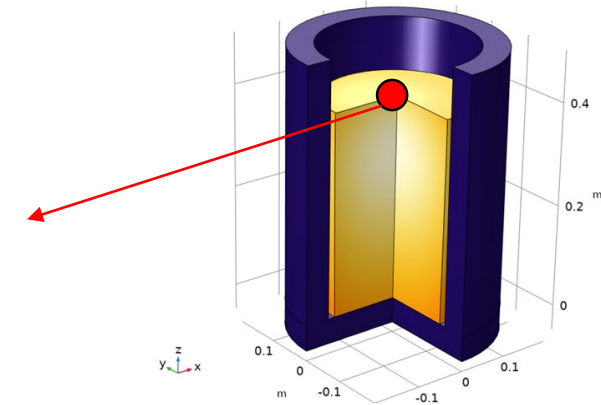
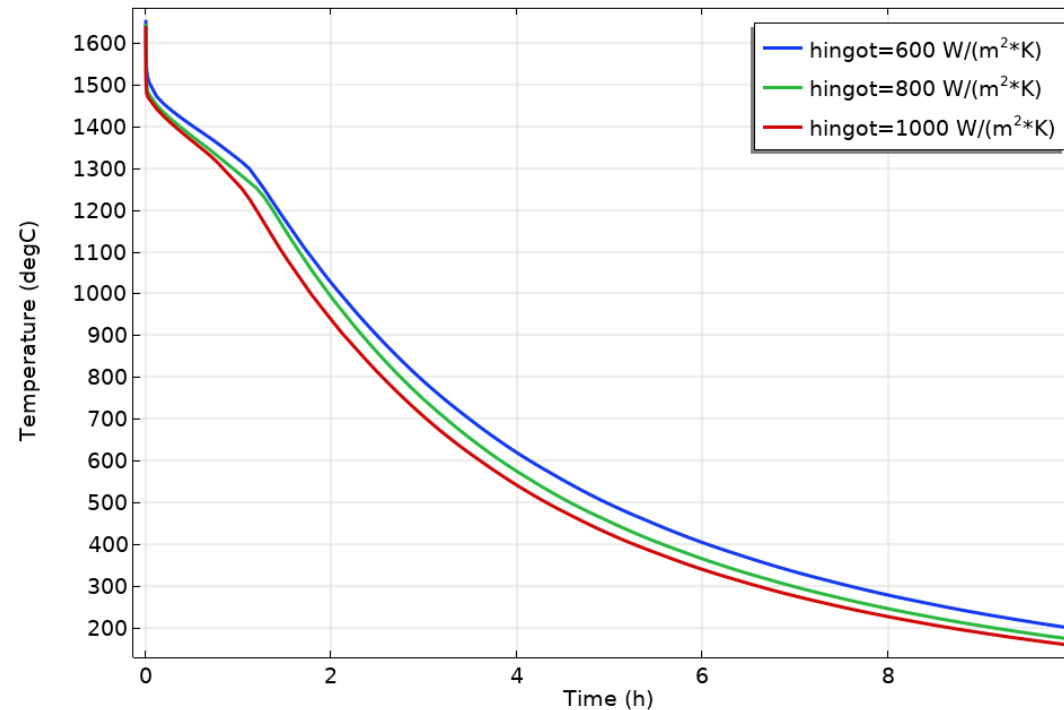
Thermo-Mechanical Results



- Von Mises stress evolution → Residual deformation and stresses prediction

Results

Influence of Lateral Conductance

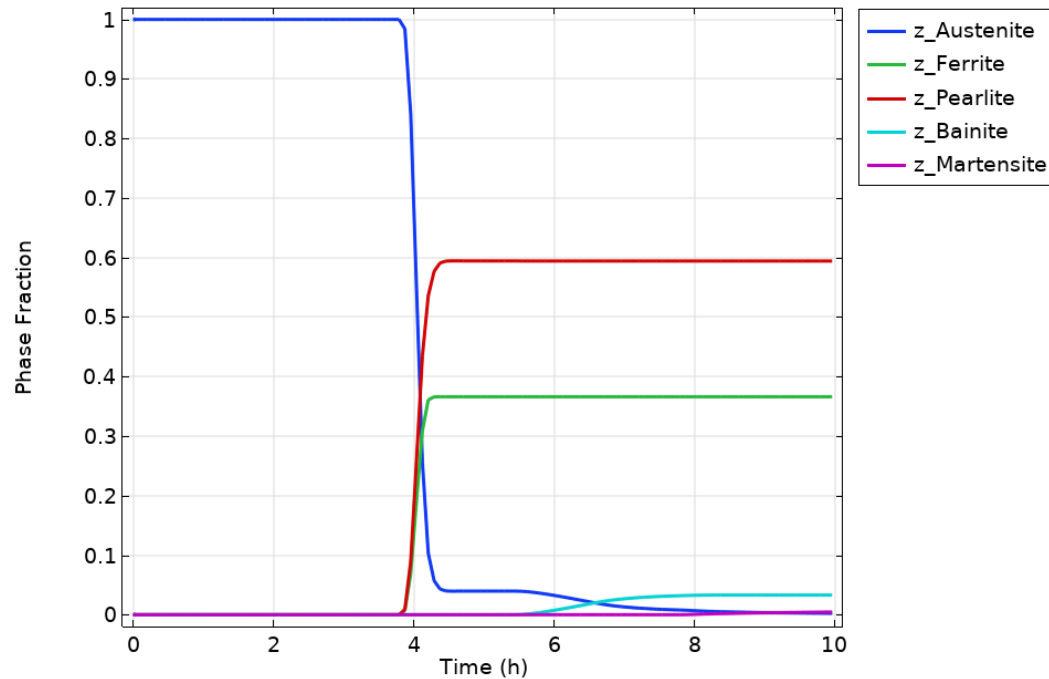


- The knowledge of the contact conditions is important to predict precisely the temperature evolution of the ingot

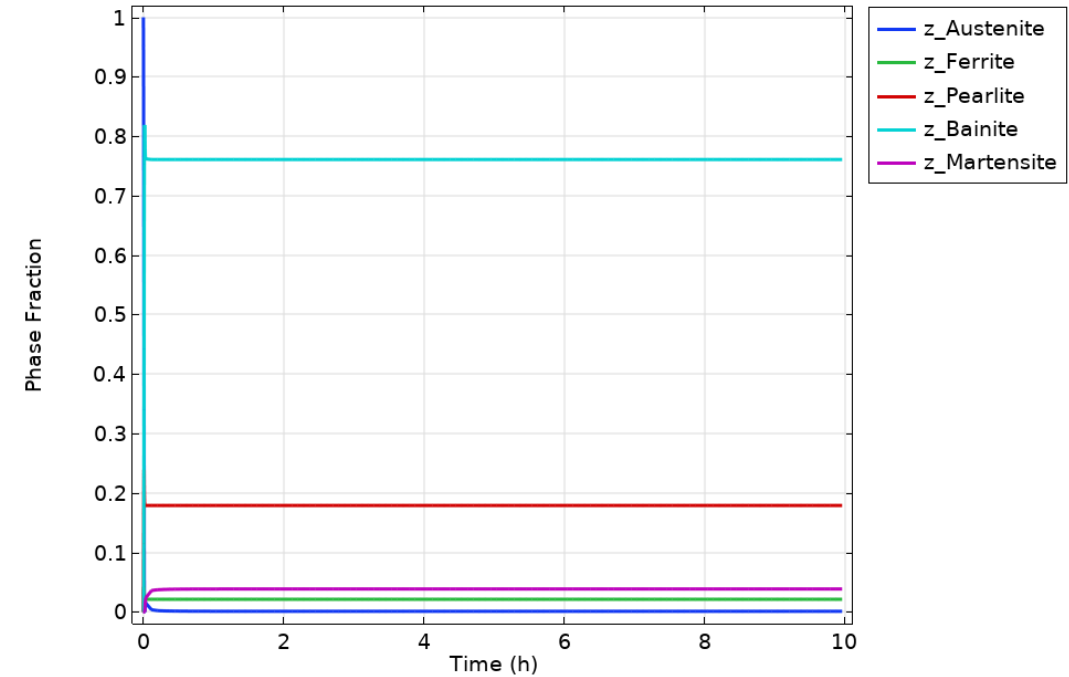
Results

Influence of Cooling conditions – Temperature Evolution

Natural Convection



Forced Convection



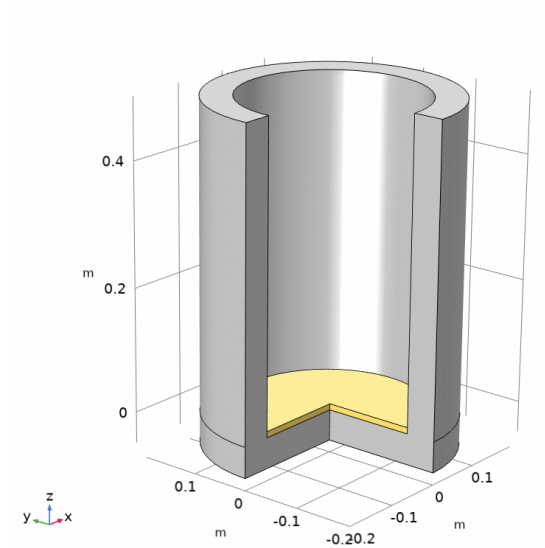
- For slow cooling (natural convection) **pearlite** is the final predominant phase
- For faster cooling condition (forced convection), **bainite** is the final predominant phase and **higher martensite** fraction

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Conclusions

Conclusions

- Development of a thermo-metallo-mechanical model
- Numerical Validation of the model
- Improvement of lateral boundary condition modeling
- Prediction of residual deformation and resulting metallurgy
- Model usable for different applications and cooling conditions



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Thank
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